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Improved High-Temperature Stability and Sun-Light-Driven Photocatalytic Activity of Sulfur-Doped Anatase TiO₂

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Of the various forms of titania (anatase, rutile, and brookite), anatase is reported to be the best photocatalyst. The anatase-to-rutile transformation in pure synthetic titania usually occurs at a temperature range of 600-700 °C. High-temperature (\geq 800 °C) stable anatase titania photocatalyst is required for antibacterial applications in building materials. A simple methodology to extend the anatase phase stability by modifying the titanium isopropoxide precursor with sulfur modification using sulfuric acid is presented. Various TTIP/H₂SO₄ molar ratios such as 1:1, 1:2, 1:4, 1:8, and 1:16 were prepared, and these samples were characterized by XRD, DSC, Raman spectroscopy, XPS, and BET surface area analysis. Sulfur-modified samples showed extended anatase phase stability up to 900 °C, while the control sample prepared under similar conditions completely converted to rutile at 800 °C. Stoichiometric modification up to 1:4 TTIP/H₂SO₄ composition (TS4) was found to be most effective in extending the anatase-to-rutile phase transformation by 200 °C compared to that of the control sample and showed 100% anatase at 800 °C and 20% anatase at 900 °C. The TS4 composition calcined at various temperatures such as 700, 800, 850 and 900 °C showed significantly higher photocatalytic activity compared to the control sample. The TS4 composition calcined at 850 °C showed visible light (sunlight) photocatalytic activity, and it decolorized the rhodamine 6G dye within 35 min (rate constant, 0.069 min⁻¹), whereas the control sample prepared under identical conditions decolorized the dye only after 3.5 h (rate constant, 0.007 min⁻¹). It was also observed that the optimal size for the highest photocatalytically active anatase crystal is \sim 15 nm. XPS studies indicated that the retention of the anatase phase at high temperatures is due to the existence of small amounts of sulfur up to 900 °C.

1. Introduction

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Nanocrystalline anatase titania has found various applications 29 in areas such as photocatalysis, photovoltaics, nanochromic display 30 devices, and self-cleaning coatings.¹⁻⁸ A high-temperature stable 31 anatase titania photocatalyst which is stable up to the curing 32 temperature of ceramic substrates is particularly desirable for 33 antibacterial material applications.^{2,9} Under normal condition, 34 anatase titania is nonreversibly converted to rutile at a temperature 35 range of 600-700 °C.8 Therefore, synthesis of a high-temperature 36 (≤800 °C) stable anatase phase is one of the major challenges in 37 the ceramic industries.⁹ Metal ion dopants have conventionally been 38 used to extend the anatase phase stability at high temperatures.^{10–14} 30 However, formation of various secondary impurity phases, such 40 as metal titanates and/or metal oxides at high temperatures, is the 41 major drawback of these methods.^{8,10} Secondary impurity phase 42 formation will reduce the photocatalytic activity and phase stability 43 of anatase titania.¹⁰ Chemical modification of the precursor by using 44 organic materials is another method used to develop high-45 temperature stable anatase titania.² This eliminates the possibility 46 of producing secondary impurity phases at high temperatures and 47

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therefore improves the required properties of titania. Various48chemical modifications have previously been reported by species49such as urea, acetic acid, and trifluoroacetic acid. 2,15,16 We have50recently found that the incorporation of nitrogen using chemical51modification of titanium isopropoxide by urea leads to the retention52of 11% of the anatase phase at 900 °C.253

Colon et al. reported the enhanced photocatalytic activity of 54 sulfated titania and sulfuric acid stabilized Cu-doped titania for 55 phenol degradation.^{17,18} The material calcined at 700 °C was crys-56 talline and contained 90% anatase, but it was completely trans-57 formed into rutile at 800 °C. Correlation between oxoanion (SO₄²⁻, 58 NO₃⁻, and PO₄³⁻) stability during calcination and photocatalytic 59 behavior of TiO2 was also reported.¹⁹ These acid treatments led to 60 an excess of adsorbed protons (Brönsted acid sites) incorporated 61 onto the TiO₂ surface. Sulfur modification in TiO₂ using hydro-62 thermal treatments were also reported.^{20,21} Ho et al. investigated 63 the antibacterial effect of S-doped titania prepared using thiourea, 64 and this study revealed that S-doped titania was effective in killing 65 Micrococcus lylae, a gram positive bacterium.^{20,21} Recent literature 66 reports indicate that sulfur could either add to the TiO₂ matrix as 67 a cation or an anion depending upon the sulfur source employed.^{22,23} 68 Umebayashi et al. reported that sulfur is doped as an anion in 69 titania.²² This anion doping shifts the absorption spectrum of titania 70 to a lower energy and effectively leads to visible light activity. 71 Ohno et al. found out that S⁴⁺ substitutes some of the lattice Ti⁴⁺, 72 leading to visible light activity.²³ Recently, visible light photoca-73

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talysis of titania doped with sulfur or nitrogen has been studied
 and reviewed by several researchers.^{24–26}

A comparative study of two different Cu precursors, CuCl₂ 76 and CuSO₄, for anatase stabilization was reported by Bokhimi 77 et al., and they suggested that 10 wt % CuSO₄ addition was 78 effective in retaining 98% anatase at 800 °C.²⁷ The stabilization 79 explained in these previous studies may not only be due to the 80 sulfate ions but is more likely due to the combined effect of 81 $Cu^{2+} \mbox{ and } SO_4{}^{2-.17,18,27}$ It should also be noted that upon 10 82 wt % addition of CuSO₄, copper segregated as bonattite 83 (CuSO₄ \cdot 3H₂O) after annealing the sample at 400 °C or a lower 84 temperature and as antierite (Cu₃SO₄(OH)₄) after annealing the 85 sample at 800 °C.27 86

Although there are a few reports available in the literature 87 on the pretreatment of titania using various inorganic acids such 88 as sulfuric acid, nitric acid, and phosphoric acid to improve 89 photocatalytic activity^{17,18,28} (using N, S, or P incorporation), 90 there have been no systematic studies reported on the develop-91 ment of a high-temperature stable anatase titania photocatalyst 92 using sulfur modification. Here, we report a simple systematic 93 94 study of the preparation of high-temperature stable anatase titania photocatalyst through modification of the precursor 95 titanium isopropoxide with various mole ratios of sulfuric acid 96 (1:1, 1:2, 1:4, 1:8, and 1:16 TTIP/H₂SO₄). The effect of sulfur 97 modification was studied by using various characterization 98 99 techniques such as XRD, Raman spectroscopy, XPS, DSC, and BET surface area analysis. This study has further investigated 100 the extent to which sulfur modification can improve the anatase 101 phase stability. It also aims to identify how the chemical 102 modification improves the photocatalytic activity of titania by 103 incorporating sulfur in a titania matrix. The novelty of the 104 current work is the development of a high-temperature stable 105 photocatalytically active anatase phase stable up to 900 °C using 106 S modification. 107

108 2. Experimental Section

2.1. Procedure. The procedure used in this study was straight-109 forward. The reagents used were titanium isopropoxide (Aldrich 110 97%) and sulfuric acid (Aldrich 95-97%). All reagents were used 111 as supplied, and no further purification was carried out. In a typical 112 experiment, 1:4 TTIP/H₂SO₄ (TS4) precursor solution and 29.7 113 mL of titanium isopropoxide (Ti(OPr)₄) waereplaced in a beaker. 114 To the above solution, 38.8 mL of concentrated H_2SO_4 (95–97%) 115 dissolved in 180 mL of water was added. The solution was then 116 stirred for 20 min and aged for 2 h at room temperature. The term 117 aging indicates the time given for the completion of a reaction. In 118 a sol-gel synthesis, this results in substantial structural reorganiza-119 tion of the gel network, which may or may not lead to the change 120 in structure and properties of the materials prepared. It was then 121 dried at 350 °C for 6 h. The dried powder was calcined at various 122 temperatures (600, 700, 800, 850, and 900 °C) at a heating rate of 123 5 °C per minute and held at these temperatures for 2 h. 124

125A similar procedure was adopted to synthesize 1:1, 1:2, 1:8,126and 1:16 samples. Samples are named as TS1, TS2, TS8, and127TS16, respectively. In all cases, the molar ratio of TTIP/H₂SO₄/128 H_2O is 1/x/100, where x corresponds to the 1, 2, 4, 8, or 16. A129control sample without any sulfuric acid was also prepared to130compare the results.

131 XRD patterns of the calcined gels were obtained with a Siemens 132 D 500 X-ray diffractometer in the range of $2\theta = 20-70^{\circ}$ using 133 Cu K α radiation. The amount of rutile in the sample was estimated 134 using the Spurr equation (eq 1).

$$F_{\rm R} = \left[\frac{1}{1 + 0.8(I_{\rm A}(101)/I_{\rm R}(110))}\right] 100 \tag{1}$$

where, $F_{\rm R}$ is the mass fraction of rutile in the sample and $I_{\rm A}(101)$ 136 and $I_{\rm R}(110)$ are the integrated main peak intensities of anatase 137 and rutile, respectively. The crystal size of the control and TS4 138 sample was calculated by using the Scherrer equation (eq 2). 139

$$\varphi = \frac{0.9\lambda}{\beta\cos\theta} \tag{2}$$

where $\varphi = \text{crystallite size}$, $\lambda = X$ -ray wavelength, $\theta = \text{Bragg}$ angle, and $\beta = \text{full width at half-maximum.}$

BET (Brunauer, Emmett, and Teller) surface area measurements and pore size analysis were carried out by nitrogen adsorption using a Quantachrome Nova 4200 surface area analyzer. The measurements were carried out at liquid nitrogen temperature after degassing the powder samples for 3 h at 300 °C.

Differential scanning calorimetry (DSC) analysis was carried out using a Rheometric Scientific DSC QC. A small amount (3 mg) of the dried precursor sample was heated from room temperature ($25 \ ^{\circ}$ C) to 650 $^{\circ}$ C at a constant heating rate of 10 $^{\circ}$ C/min.

X-ray photoelectron spectroscopy (XPS) analyses were performed using a Thermo VG Scientific Sigma Probe spectrometer. The instrument employs a monochromated Al Ka X-ray source ($h\nu = 1486.6$ eV), which was used at 140 W. The area of analysis was approximately 500 μ m in diameter for the samples analyzed. For survey spectra, a pass energy of 100 eV and a 0.4 eV step size were employed. For C 1s and Ti 2p high-resolution spectra, a pass energy of 20 eV and a 0.1 eV step size were used. For O 1s high-resolution spectra, a pass energy of 20 eV and a 0.2 eV step size were used. For S 2p high-resolution spectra, a pass energy of 50 eV and a step size of 0.2 eV were employed. Charge compensation was achieved using a low-energy electron flood gun. Quantitative surface chemical analyses were calculated from the high resolution core level spectra, following the removal of a nonlinear Shirley background.

2.2. Photocatalytic Study. In a typical experiment, 0.06 g 170 of TS4 sample calcined at 850 °C was added to an aqueous 171 solution of rhodamine 6G solution (50 mL, 5×10^{-6} M), and 172 this suspension was irradiated in a Q-Sun Xenon solar simulator 173 chamber (Intensity max., 0.68 W/m² at λ max. of 340 nm) with 174 continuous and uniform stirring. The degradation of rhodamine 175 dye was monitored by taking 4 mL aliquots at different intervals 176 of time. These aliquots were centrifuged, and absorption spectra 177 of the samples were recorded using as UV/vis spectrophotom-178 eter. Similar experiments were carried out for the control sample 179 calcined at 850 °C. The sunlight-driven photocatalytic activity 180 of the TS4 and control titania calcined at 850 °C was determined 181 by carrying out the reaction under Dublin sunlight (October 27, 182 2007). The intensity of the sunlight was determined by a Solar 183 Light Co. broad-band radiometer PMA 2107 (Philadelphia), 184 which gave approximately 12.0 W/m². The rate of degradation 185 was assumed to obey pseudo-first-order kinetics, and hence, the 186 rate constant for degradation, k, was obtained from the first-187 order plot (eq 3). 188

$$\ln\left(\frac{A_0}{A}\right) = kt \tag{3}$$

where A_0 is the initial absorbance, A is the absorbance after a 190 time (t), and k is the pseudo-first-order rate constant. 191

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Improved Stability of Anatase TiO₂

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Figure 1. XRD spectra of calcined sample at 700 and 800 °C; (a) control, (b) TS4.



Figure 2. XRD spectra of TS4 sample calcined at (a) 850 and (b) 900 $^{\circ}$ C.

3. Results

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3.1. XRD Analysis. The anatase-to-rutile transformation of 193 the calcined samples was studied by the powder X-ray diffrac-194 tion (XRD) technique. The XRD patterns of the samples 195 calcined at 700, 800, 850, and 900 °C are shown in Figures 1 196 and 2. The weight fraction of the anatase retained in the sample 197 after various calcination temperatures was calculated by com-198 paring the XRD-integrated intensities of the (101) reflection of 199 anatase and the (110) reflection of rutile. It was found that 100% 200 anatase TiO₂ was observed at the temperature of 800 °C for 201 the sample prepared with or above TS4 composition. In 202 comparison, it was found that even at 700 °C, the control sample 203 showed 82% rutile, while in the case of the samples prepared 204 205 with lower mole ratios of H_2SO_4 (TS1 and TS2), the anataseto-rutile transformation commenced above 700 °C, and complete 206 transformation to rutile occurred only at 900 °C (Figure 3). 207 There is no notable influence on the anatase-to-rutile transfor-208 mation temperature above TS4 modification. The TS8 and TS16 209 samples show similar phase distribution behavior up to 800 °C 210 and were found to be slightly less effective in retaining the 211 anatase phase at 900 °C compared to the TS4 sample. The 212 optimum level of modification was found in the TS4 sample. 213 The control sample was fully converted to rutile at 800 °C 214 (Figure 1), while the TS4 sample retained 100% anatase at 800 215 °C, 98% anatase at 850 °C, and 20% anatase even at 900 °C 216 (Figure 2 and 3). 217



Figure 3. Anatase content in the control and various sulfur-modified samples (from XRD data) at different temperatures (error = $\pm 5\%$).

3.2. Raman Studies. Raman spectroscopy was applied as 218 an additional characterization tool to confirm the phase stability 219 of titania at high temperatures. According to factor group 220 analysis, the anatase phase consists of five Raman-active modes, 221 while the rutile phase consists of four²⁹ (i.e., anatase: 144, 197, 222 395, 513, and 639 cm⁻¹; rutile: 143, 233, 447, and 610 cm⁻¹). 223 Figure 4 indicates the Raman spectra of the control and TS4-224 modified sample calcined at 800 °C. All of the peaks present 225 in the control sample were due to the rutile phase, and in the 226 TS4 sample, all of the peaks were due to the anatase phase. 227 The TS4 sample calcined at 900 °C (Figure 5) showed three 228 anatase peaks which correspond to the amount of anatase phase 229 present in the 900 °C sample. These results indicated that Raman 230 spectroscopy studies were consistent with the XRD results. 231

3.3. XPS Analysis. XPS measurements were carried out to 232 investigate the incorporation of sulfur in the titania matrix. It 233 was previously reported that a sulfur-containing material shows 234 a binding energy value of around 170 eV.18 The high-resolution 235 XPS spectra of the S_{2p} region of the TS4 sample calcined at 236 different temperatures, 700, 800, and 900 °C, are displayed in 237 Figure 6. The presence of sulfur, carbon, oxygen, and titania in 238 the samples was confirmed by XPS analysis. The presence of 239 sulfur was confirmed by a peak at 168.4 eV (Figure 6). A 240 previous literature report indicated that the peak at 168.4 eV is 241 due to the presence of the S6+ cation.^{21,23b} In all S-modified 242 samples, a peak at 168.4 eV was visible. In Figure 6a, the TS4 243

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Figure 4. Raman spectra of the samples at 800 °C; (a) control, (b) TS4.

sample calcined at 700 °C shows a sharp peak at 168.4 eV, but 244 at higher temperatures of 800 and 900 °C (Figure 6b and c), 245 the peak becomes broad. This peak broadening observed in the 246 high-binding energy side of the S peak (168.4 eV) can be 247 attributed to two components, $S_2p_{3/2}$ and $S_2p_{1/2}$. These peaks 248 249 are only ~ 1.1 eV apart and thus cannot be resolved from one another. Typically, the S₂p_{3/2} peak at lower binding energy 250 is about twice the intensity or area of the higher binding energy 251 $S_2p_{1/2}$ peak. The XPS spectra of TS4 samples at 800 and 900 252 °C (Figure 6b and c) show a small amount of peak splitting, 253 which indicates that at higher temperature, there may be some 254 amount of S⁴⁺ cation also present.^{23b} If the S concentration was 255 higher, S⁶⁺ and S⁴⁺ peaks could be clearly resolved. However, 256 as the sulfur amount present in the samples is less than 2 atom 257 %, peak resolution is difficult. Table 1 shows that the presence 258 of sulfur gradually reduces as the calcination temperature 259 increases. 260

3.4. BET Surface Area Analysis. The BET surface area of 261 the TS4 and control titania samples calcined at 700, 800, and 262 900 °C are listed in Table 1. The TS4 sample calcined at 700 263 °C showed a type-IV isotherm (Figure 7). The results indicated 264 265 that the surface area of the samples was strongly dependent on the thermal treatment. The surface area of the TS4 sample 266 decreased from 28 to 6 m²/g as the calcination temperature 267 increased from 700 to 900 °C. BET analysis of the TS4 sample 268 calcined at higher temperatures showed a larger surface area 269 (18 m²/g at 800 °C and 6 m²/g at 900 °C) compared to that of 270 the control sample calcined at the same temperature (6.4 m^2/g 271 at 800 °C and 0.96 m²/g value at 900 °C). Therefore, the sulfur 272 modification was found to increase the surface area of titania 273 significantly. 274



Figure 5. Raman spectra of the samples at 900 $^{\circ}$ C; (a) control, (b) TS4.

3.5. Photocatalytic Study. In an attempt to assess the 275 performance of the high-temperature stable anatase titania, the 276 photocatalytic activity of the TS4 sample calcined at 850 °C 277 and a similar control sample (Figure 8) was analyzed via a 278 rhodamine 6G degradation. The control sample degraded the 279 dye within 80 min, whereas the S-modified sample completely 280 degraded the dye within 12 min. The control and the TS4 sample 281 calcined at 700, 800, and 900 °C were also subjected to 282 photocatalytic study (Supporting Information 1). The TS4 283 sample calcined at 800 °C degraded the rhodamine dye within 284 15 min, whereas the control (800 °C) required 40 min. This 285 enhanced efficiency of the S-modified titania photocatalyst is 286 reflected in kinetic analysis of the results (Supporting Informa-287 tion 1 and 2). The rhodamine degradation follows pseudo-first-288 order kinetics. First-order degradation rate constants, obtained 289 by plotting the natural logarithm of the absorbance against 290 irradiation time for the control and TS4 sample were calculated. 291 The rate constant values recorded for the TS4 samples calcined 292 at various temperatures were consistently and significantly 293 superior to those recorded for the control samples under identical 294 conditions (Supporting Information 1 and 2). Dark studies, 295 where the above experiments were repeated in the absence of 296 light, showed that there were no adsorption effects. A sample 297 left for 24 h showed little change in absorbance of the rhodamine 298 dye. The visible light (sunlight) activity of the S-doped materials 299 has been examined. The results of sunlight photocatalytic 300 activity are shown in Figure 9. The TS4 composition calcined 301 at 850 °C decolorized the rhodamine 6G dye within 35 min 302 (rate constant, 0.069 min^{-1}), whereas the control sample 303 prepared under identical conditions decolorized the dye after 304



Figure 6. XPS spectra of TS4 samples calcined at (a) 700, (b) 800, and (c) 900 °C.

 TABLE 1: Surface Area of the TS4 and Control Titania at Different Temperatures

sample	BET surface area (m ² /g) (±2.5%)	pore volume (cc/g)	pore diameter (nm)	sulfur content (atom %) by XPS (±5%)
TS4 800 °C	18.7	0.22	2.8	0.99
TS4 700 °C	28.1	0.37	44.1	1.24
TS4 900 °C	6.1	0.04	1.0	0.84
control 700 °C	25.6	0.16	14.2	0
control 800 °C	6.4	0.06	1.5	0
control 900 °C	1.0	0.00	1.0	0

305 3.5 h (rate constant, 0.007 min^{-1}). The rate constant values show 306 that sulfur-doped titania has 10-fold greater photocatalytic 307 activity compared to that of the control sample.

4. Discussion

4.1. Anatase Phase Stability. Titanium isopropoxide reacts 309 with water, and an aquo complex (Ti-OH₂)⁴⁺ is initially 310 formed, which decomposed immediately to produce titanium 311 hydroxo (Ti-OH)³⁺/oxo (Ti=O)²⁺ complex precipitates.³³ In 312 order to avoid this faster hydrolysis (formation of the Ti-O 313 bond) and condensation (removal of water or alcohol after the 314 formation of the Ti-O bond), chemical additives (complexing 315 molecules) are added to moderate the rate of the reaction.³⁴ This 316 usually leads to the formation of a transparent sol instead of 317 precipitating oxo/hydroxo complexes.33,35,36 The hydrolysis and 318 condensation of titanium isopropoxide can be effectively 319 controlled at a low pH.37,38 During the hydrolysis in the presence 320 of an acid, the OR groups attached to the metal are protonated 321

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Figure 7. Type-IV isotherm of the sample calcined at 700 °C; (a) control, (b) TS4.

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Figure 8. Kinetic study and absorption spectra of rhodamine dye degradation using (a) the control and (b) TS4 samples calcined at 850 °C under UV light. A_0 is the initial absorbance, and A is the absorbance after a time for the rhodamine dye degradation (error = ±10%).

first by the H_3O^+ , which causes the OR group to reduce its 322 electronegativity (its charge becomes more positive).^{37,38} The 323 metal ion, which is already positively charged due to the 324 electropositive nature, begins to repel the protonated OR. This 325 repulsion between protonated species will decrease their interac-326 tion, which causes a rate of decrease in the condensation; hence, 327 the rate of hydrolysis also decreases. In the present study, 328 329 sulfuric acid was used as a chemical modifier; titanium isopropoxide reacted exothermally with sulfuric acid to form 330 titanyl oxysulfate and thus decomposed at higher temperature 331 332 to produce sulfur-doped titania, as indicated by eqs 4 and 5.

$$Ti(OPr)_4 + H_2SO_4 + H_2O -$$

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$$TiOSO_4 + 4(CH_3 - CH(OH) - CH_3)$$
 (4)

$$2\text{TiOSO}_4 \rightarrow 2\text{Ti}_{1-x}\text{S}_x\text{O}_2 + (2-x)\text{SO}_2 + \text{O}_2 \qquad (5)$$

The titanyl oxysulfate is stable up to a temperature of 600 °C, 336 as indicated by the XRD (JCPDS file no. 81-1566) (Figure 10). 337 DSC analysis of the sulfur-modified sample (TS4 sample) 338 showed an endothermic peak at 620 °C (Figure 11) due to the 339 decomposition of TiOSO₄ (eq 5). This is consistent with the 340 XRD pattern at 700 °C (Figure 1), where the TS4 sample 341 showed pure anatase phase only. This shows that above 600 342 343 °C, TiOSO₄ began to decompose and formed the sulfur-doped anatase phase. The TiOSO₄, which is stable up to 600 °C, 344 indicated by the XRD and DSC results (Figures 10 and 11), 345 has a basic structure (Scheme 1a) consisting of connected 346 [TiO₆]²⁻ octahedra and [SO₄]²⁻ tetrahedra.³⁹ Upon further heat 347 treatment of the TS4 sample above 600 °C, the decomposition 348 of TiOSO₄ occurs (eq 5) to form the edge-shared anatase titania, 349

as shown in Scheme 1b. The final conversion to the thermo-350 dynamically stable rutile phase occurs at higher temperatures 351 (900 °C), as illustrated in Scheme 1c. It has been reported that 352 sulfur doping is difficult to achieve in titania due to the larger 353 formation energy needed to substitute the oxygen by sulfur to 354 form $TiO_{2-x}S_x$.^{40,41} Therefore, it is important to note that the 355 TS4 sample contained 1.24, 0.99, and 0.84 atom % sulfur at 356 700, 800, and 900 °C, respectively (Table 1). Thus sulfur doping 357 was easily achieved by the decomposition of TiOSO₄ formed 358 by the reaction between titanium isopropoxide and sulfuric acid 359 (eq 5). 360

Anatase (tetragonal, a = b = 3.78 Å; c = 9.50 Å) and rutile 361 (tetragonal, a = b = 4.58 Å; c = 2.95 Å) are the two main 362 crystalline forms of titania. It can also exists in brookite form 363 (rhombohedral, a = 5.43 Å; b = 9.16 Å; c = 5.13 Å), although 364 this particular phase did not occur at any stage of this study. 365 Even though anatase and rutile exist in a tetragonal structure, 366 experimental evidence indicates that anatase is more stable 367 kinetically than rutile at room temperature and atmospheric 368 pressure.^{2,37} Rutile is reported to be more stable thermodynami-369 cally than anatase at normal temperature.37,42 The thermody-370 namic phase stability calculation by Zhang and Banfield shows 371 anatase titania transforms to rutile only after growing to a 372 crystallite size of ~ 14 nm.⁴²⁻⁴⁴ Below this critical size of ~ 14 373 nm, the anatase phase is more stable. The crystallite sizes 374 calculated by using the Scherrer equation (eq 3) for the TS4 375 sample and control sample are shown in the Table 2. Here, in 376 the case of the control titania mentioned above, the critical size 377 is already exceeded at 700 °C (17.7 nm) itself, whereas the TS4 378 sample reached this limit (15.7 nm) only after the calcination 379



Figure 9. Kinetic study and absorption spectra of rhodamine dye degradation using (a) the control and (b) TS4 samples calcined at 850 °C under sunlight. A_0 is the initial absorbance, and A is the absorbance after a time for the rhodamine dye degradation (error = $\pm 10\%$).



Figure 10. XRD of the samples calcined at 600 °C; (a) control, (b) TS4

0.0 b -0.2 Heat flow (millical/sec) a -0.4 -0.6 -0.8 -1.0 620 °C -1.2 100 200 300 400 500 600 0 Temperature(°C)

Figure 11. DSC curve of (a) the control and (b) TS4.

molar concentration of sulfuric acid showed that S modification of TiO₂ was most effective at four times the concentration of the titanium isopropoxide (TS4 sample). Studies of the phase transformation temperatures by XRD and Raman spectroscopy support this conclusion. During the calcination process, surfaceadsorbed water, hydroxyl groups, and bridged OH groups attached to TiO₂ will be removed. This elimination causes the formation of an oxygen vacancy in the titania matrix.^{17,18} In these oxygen vacancy sites, the sulfur may be doped as an anion/ cation, but it has been previously reported by Yu et al. that

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at 850 °C. This suggests that the anatase phase is stabilized as 380 381 382 383 384 385 386 387

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a result of sulfur doping due to the decrease in crystallite size. It has been reported that sulfuric acid pretreatment on TiO₂ particles provided 90% anatase at 700 °C but was completely transformed to rutile phase at 800 °C.18,19 In the present investigation, it appears that the S modification of TiO₂ by

sulfuric acid was effective in extending the anatase stability up to 900 °C. Although the lower mole percentage samples TS1 and TS2 improved the anatase stability, gradual increments in

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SCHEME 1: Schematic Representation of Decomposition of TiOSO₄ to Anatase and Its Transformation to Rutile (a: TiOSO₄; b: Anatase Titania; c: Rutile Titania)



TABLE 2: Crystallite Size of Anatase

	Cryst	Crystallite size of anatase (nm) $(\pm 5\%)$				
samples	700 °C	800 °C	850 °C	900 °C		
control TS4	17.7 7.09	^а 12.9	^а 15.7	^a 23.6		

^a Anatase phase is transformed to rutile.

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anionic sulfur doping is difficult to achieve because the ionic radius of the S²⁻ (1.7 Å) ion is larger than that of O²⁻ (1.22 Å).²¹ The bond strength of the Ti-S bond (418.0 kJ/mol) is less than the already existing Ti-O bond (672.4 kJ/mol).³⁰ Therefore, the substitution of Ti⁴⁺ by S⁶⁺ is chemically more favorable than the replacement of O²⁻ by S²⁻. In this study, the presence of S⁶⁺ ions (the oxidation state of sulfur in sulfuric acid is +6) in the titania lattice was confirmed by XPS measurement, which showed a peak at 168.4 eV (Figure 5).

408 **4.2. Photocatalysis.** Nanocrystalline titania exhibits photo-409 catalytic activity in the presence of UV light, and it can 410 decompose organic pollutants. This activity depends upon 411 several factors, such as the rate of electron—hole recombination, 412 the number of electrons created, the phase composition (anatase 413 or rutile), the surface area, the crystallinity, as well as the 414 crystallite size of the TiO₂ and the adsorption properties of the

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dyes on the surface of the TiO2 used.29 The sulfur-modified 415 sample exhibited higher photocatalytic activity than the control 416 sample prepared under identical conditions. The photocatalytic 417 activity of the control sample decreased with increased calcina-418 tion temperature, whereas the sulfur-modified sample showed 419 improvement in photocatalytic activity as the calcination tem-420 perature increased up to 850 °C (Supporting Information 1 and 421 2). It has been reported that the adsorption affinity of the anatase 422 phase for the organic compound is higher than that of rutile, 423 and anatase exhibits lower rates of recombination in comparison 424 with those for rutile due to its 10-fold greater rate of electron 425 trapping.⁴⁵ In this particular study, the anatase phase stability 426 of the nanocrystalline TiO₂ powders at high calcination tem-427 peratures due to the sulfur doping is the main reason for the 428 high photocatalytic activity. Sulfur doping leads to three main 429 improvements in the modified sample. First, sulfur doping 430 significantly increased the surface area of the high-temperature 431 stable anatase titania (Table 1), which leads to the improved 432 adsorption and thereby higher concentration of the reactant 433 rhodamine dye molecules near the active centers of titania. 434 Second it significantly changed the band gap of high-temperature 435 stable anatase titania (Supporting Information 3 and 4). Finally, 436 it significantly reduced the crystallite size of the high-temper-437 ature stable anatase titania compared to that of the control 438 samples (Table 2). Normally the surface of the TiO₂ is positively 439 charged in acidic media and negatively charged in basic 440 media.46,47 Thus, here, the rhodamine 6G dye tends to adsorb 441 on the surface of the TiO₂ by the negatively charged carbonyl 442 group of the dye. Both UV and visible light irradiation show a 443 gradual decrease in absorption of the dye without any blue shift 444 in absorption maxima (Figures 8 and 9). It has been reported 445 that the blue shift in the absorption maxima represents the dye 446 degradation pathway via the elimination of the diethyl group.^{46,48} 447 Therefore, it has been possible to conclude here in our study 448 that the active radical species (hydroxyl and super oxide radicals) 449 generated during UV and visible light illumination attack the 450 ring structure (chromophore) to induce a cleavage rather than 451 attacking the diethyl amino group (auxochrome) to direct the 452 complete degradation of the dye. Also, it should be noted that 453 there are no extra peaks appearing in UV-visible spectra 454 (Figures 8 and 9), which leads to the conclusion that the dye is 455 not photobleached during the photocatalytic reaction. 456

The photocatalytic activity of the TS4 samples increased with 457 an increase in calcination temperature (Supporting Information 458 1 and 2), and it decreased after 850 °C. The reason for the 459 highest activity of the TS4 sample calcined at 850 °C can be 460 explained by the critical crystallite size limit of the photocatalytic 461 activity. It is well-known that the photocatalytic activity of 462 nanocrystalline titania is strongly dependent on its crystallite 463 size.^{44,49,50} The reported critical crystallite size limit of TiO₂ 464 for better photocatalytic activity of nanocrystalline titania is ~ 15 465 nm. Above or below this value, the photocatalytic activity 466 decreases.^{41,42} For a better photocatalytic performance, the rate 467 of the surface charge carrier recombination process should be 468 minimum, but the interfacial charge-transfer process should be 469 maximum.44 As the nanocrystalline size decreases, the number 470 of active surface sites which enhance the rate of the interfacial 471 charge-transfer process increases; hence, the photocatalytic 472 activity also increases. However, after reaching a critical size 473 $(\sim 15 \text{ nm})$, the surface charge recombination dominates; hence, 474 it reduces the photocatalytic activity. In our case, the sulfur-475 modified sample reached this limit at 850 °C, which shows the 476 highest activity among the various TS4 samples calcined at 477 different temperatures. 478 ohio1/yjy-yjy/yjy-yjy/yjy99907/yjy5624d07z | xppws | 23:ver.3 | 4/3/08 | 12:34 | Msc: jp-2007-074847 | TEID: mlk00 | BATID: jp5d216 |

Improved Stability of Anatase TiO₂

479 **5.** Conclusions

High-temperature stable anatase (up to 20% weight fraction) 480 titania, existing up to 900 °C, has been successfully prepared 481 by chemically modifying the precursor with sulfuric acid. This 482 method eliminates the formation of various secondary impurity 483 phases such as metal titanates and/or metal oxides at high 484 temperatures, which can lead to the reduction of the photocata-485 lytic activity and phase stability of anatase titania. Sulfur doping 486 in high-temperature stabilized anatase titania was easily achieved 487 by the decomposition of TiOSO₄ formed by the initial reaction 488 between titanium isopropoxide and sulfuric acid. The high-489 temperature stability of anatase titania and sulfur doping was 490 investigated by different characterization techniques such as 491 XRD, Raman spectroscopy, DSC, XPS, and BET surface area 492 493 analysis. All of the sulfur-modified samples showed significantly 494 higher photocatalytic activity compared to the control titania. The enhanced efficiency of the sulfur-modified titania photo-495 catalyst is reflected in kinetic analysis. The pseudo-first-order 496 rates for the oxidation of rhodamine dye by the sulfur-modified 497 sample calcined at different temperature (700, 800, 850, and 498 900 °C) are significantly higher than those for the corresponding 499 control samples. The TS4 composition calcined at 850 °C 500 showed the visible light photocatalytic activity, and it decol-501 orized the rhodamine 6G dye within 35 min (rate constant 0.069 502 503 \min^{-1}), whereas the control sample prepared under identical conditions decolorized the dye after 3.5 h (rate constant 0.007 504 \min^{-1}). 505

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511 **Supporting Information Available:** Rate constant and 512 photocatalytic activity data and spectra and diffuse reflectance 513 spectra. This material is available free of charge via the Internet 514 at http://pubs.acs.org.

515 **References and Notes**

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