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# Incorporation of Fibrin Into a Collagen–Glycosaminoglycan Matrix Results in a Scaffold With Improved Mechanical Properties and Enhanced Capacity to Resist Cell-Mediated Contraction

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### Accepted Manuscript

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### Incorporation of fibrin into a collagen-glycosaminoglycan matrix results in a scaffold with improved mechanical properties and enhanced capacity to resist cell-mediated contraction.

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#### Abstract:

Fibrin has many uses as a tissue engineering scaffold, however many *in vivo* studies have shown a reduction in function resulting from the susceptibility of fibrin to cell-mediated contraction. The overall aim of the present study was to develop and characterise a reinforced natural scaffold using fibrin, collagen and glycosaminoglycan (FCG), and to examine the cell-mediated contraction of this scaffold in comparison to fibrin gels. Through the use of an injection loading technique, a homogenous FCG scaffold was developed. Mechanical testing showed a six-fold increase in compressive modulus and a thirty-fold increase in tensile modulus of fibrin when reinforced with a collagen-glycosaminoglycan backbone structure. Human vascular smooth muscle cells (vSMCs) were successfully incorporated into the FCG scaffold and demonstrated excellent viability over 7 days, while proliferation of these cells also increased significantly. VSMCs were seeded into both FCG and fibrin-only gels at the same seeding density for 7 days and while FCG scaffolds did not demonstrate a reduction in size, fibrin-only gels contracted to 10% of their original diameter. The FCG scaffold, which is composed of natural biomaterials, shows potential for use in applications where dimensional stability is crucial to the functionality of the tissue.

#### Keywords:

Fibrin

**Tissue Engineering** 

Cardiovascular scaffold

Heart valve

#### 1. Introduction

Scaffolds for tissue engineering (TE) provide a template for cells to lay down extracellular matrix (ECM). Biocompatibility and ease of sterilisation are basic scaffold requirements, while scaffold geometry, degradation kinetics and mechanical properties each play a key role in the generation of the tissue construct in vitro (Jana et al. 2014; Li et al. 2014). There are unique challenges associated with the design of scaffolds for applications where the shape and dimensional stability of the scaffold is critical to its function. For example, heart valve constructs that have undergone a period of in vitro remodelling must be able to withstand the challenging haemodynamic environment at the donor site from the moment of implantation. If the construct does not function correctly, i.e. with full coaptation of the leaflets, it will not be able to sustain its role as a valve over the long term. To date, a variety of scaffolds have been proposed for heart valve TE, using both synthetic (Hoerstrup et al. 2002; Ramaswamy et al. 2010; Sodian et al. 2000; Sant et al. 2013; Kalfa et al. 2010; Schmidt et al. 2010; Mol et al. 2006) and biological (Flanagan et al. 2009; Yamanami et al. 2010; Weber et al. 2013; Syedain et al. 2013) materials and these have demonstrated varying degrees of success once implanted in vivo. Of these options, we believe that fully biological heart valves hold greater promise than synthetic or decellularised valves for the development of long-term heart valve replacements which will have the ability to grow and remodel with the patient.

Fibrin, due to its inherent biocompatibility and plasmin-controlled remodelling (Collen 2001), shows particular promise. Fibrin is a naturally-occurring polymer involved in the wound-healing response. It can be extracted from the blood of a potential patient, creating a biocompatible, autologous material, whose peptide chains and integrin binding sites encourage cell adhesion, migration, proliferation, and angiogenesis (Haisch et al. 2000; Jockenhoevel et al. 2001; Weinandy et al. 2014). Fibrin can be used to encapsulate cells, creating a construct with a homogenous cell distribution and has been used in a range of applications including neural regeneration, wound healing, bone grafts, cartilage repair and cardiovascular applications (Ahmed et al. 2008). The rate of fibrin degradation can be controlled through the use of protease inhibitors (Cholewinski et al. 2009; Eyrich et al. 2007; Collen 2001) and fibrin also encourages ECM deposition and retention by the cells that it encapsulates (Ahmann et al. 2010; Ameer et al. 2002; Ye, Zünd, Benedikt, et al. 2000). Conversely, the weak mechanical properties of fibrin and susceptibility to cell-mediated contraction mean that sustaining dimensional stability is very difficult when using fibrin as a lone scaffold material. Many in vivo studies have shown a reduction in function resulting from this cell-mediated contraction (Flanagan et al. 2009; Syedain et al. 2011; Weber et al. 2013). In light of this, fibrin has been incorporated in combination with biomaterials such as hyaluronic acid (Lee & Kurisawa 2013), polyethylene oxide (Akpalo et al. 2011) and alginate

(Shikanov et al. 2009) in interpenetrating networks. Fibrin has also been combined with collagen in mixed gels (Cummings et al. 2004), with a limited increase in mechanical properties reported. Additionally, fibrin has been used with porous, synthetic scaffolds (Gundy et al. 2008; Lesman et al. 2011; Mol et al. 2005; Moreira et al. 2014) and while this approach has improved the structural integrity of fibrin, acidic by-products are released during the subsequent degradation of the synthetic scaffold (Sung et al. 2004; Monfoulet et al. 2014; Busa & Nuccitelli 1984).

The use of natural materials to reinforce a fibrin gel, with the aim of minimising cellmediated contraction of the fibrin gel, has not been examined in detail. Moreover, the literature shows that a stiffer scaffold has a greater resistance to cell-mediated contraction (Sheu et al. 2001; Keogh et al. 2010). Our research group has developed a range of collagenbased structures, with interconnected, homogenous pores and a dehydrated macrostructure for a variety of applications, from bone grafts to gene delivery (O'Brien et al. 2005; O'Brien et al. 2004; Keogh et al. 2010; Cunniffe et al. 2010; Roche et al. 2014; Tierney et al. 2012). The mechanical properties of the freeze-dried collagen can be customised using different manufacturing methods and constituents, for example, the addition of glycosaminoglycans (GAGs) has been shown to increase the stiffness of the collagen matrix (Gleeson et al. 2010; Haugh et al. 2011; O'Brien et al. 2004). Consequently, the present study hypothesises that a fibrin gel that is reinforced with a freeze-dried, collagen glycosaminoglycan (CG) structure will provide a more dimensionally stable scaffold for tissue engineering applications, by resisting cell-mediated contraction.

Hence, the overall aim of this study was to develop a fibrin-collagen-glycosaminoglycan (FCG) scaffold that could resist cell-mediated contraction and provide a dimensionally stable structure for tissue engineering applications. The specific objectives were (i) to develop methods of combining a fibrin gel and a CG matrix and to characterise the microstructure of the resultant FCG scaffolds; (ii) to investigate the tensile and compressive moduli of fibrin when reinforced with a CG matrix; (iii) to investigate the response of human vascular smooth muscle cells (vSMCs) to the FCG scaffold, and (iv) to examine the cell-mediated contraction of FCG scaffolds in comparison to fibrin gels alone.

#### 2 Materials and Methods

#### 2.1 Scaffold Fabrication

#### 2.1.1 Fabrication and crosslinking of CG structure

CG sheets were fabricated, as previously described, using a lyophilisation process (O'Brien et al. 2005) with a collagen (type 1 bovine) (Integra Life Science, Plainsboro, NJ, USA) concentration of 0.75% w/v solution and a GAG (chondroitin sulphate) (Sigma-Aldrich, Arklow, Ireland) concentration of 0.044% w/v (Tierney et al. 2009). The concentrations used were based on previous characterisation work performed within the laboratory. The CG

suspension was freeze-dried to a final freezing temperature of -10°C using a freezing rate of 1°/min, followed by subsequent drying, to produce a homogenous structure with an average pore size of 150µm and porosity of 99% (Haugh et al. 2010). Dehydrothermal treatment using a vacuum oven (Vacucell, MMM Group, Munich, Germany) at 0.05 bar and 105°C, over 24 hours, was used to physically crosslink and sterilise the CG matrix (Haugh et al. 2009). Discs (15.8mm diameter, 2mm high) of the sterile CG matrix were cut using a biopsy These were chemically crosslinked using 1-ethyl-3-(3punch. dimethylaminopropyl)carbodiimide (EDAC) in combination with N-hydroxysuccinimide (NHS) as described previously (Haugh et al. 2011), with the difference of using ethanol as the solvent to further increase stiffness (Barnes et al. 2007). Discs were washed in decreasing concentrations of ethanol and stored in phosphate buffered saline (PBS) (Sigma-Aldrich) prior to use, which was always within 12 hours.

#### 2.1.2 Incorporation of fibrin into CG matrix

Fibrin gels were fabricated as previously described (Jockenhoevel et al. 2001). Briefly, bovine fibrinogen (Sigma-Aldrich) was dissolved in PBS and dialysed against 4L of tris-buffered saline (TBS) in 6,000- 8,000MW dialysis tubing (Spectrum Labs, Breda, The Netherlands) overnight. After sterile filtration, the fibrinogen concentration was quantified by measuring the absorbance at 280nm using a spectrophotometer (Nanodrop 2000; Thermo Fisher Scientific, Delaware, USA). The final concentration of fibrinogen solution was adjusted to 10mg/ml with PBS. To make fibrin gels of 1000µl, 75µl of 0.05M CaCl<sub>2</sub>, 350µl TBS and 75µl of 40IU/ml of thrombin (Sigma-Aldrich), were pipetted into a 24-well plate into which 500µl of the 10mg/ml fibrinogen solution was added for subsequent polymerisation. After transferring the fibrin to an incubator at 37°C for 60 minutes, polymerisation was complete. To fabricate gels of different sizes, these ratios were maintained and the volume changed.

In order to fabricate the fibrin-collagen-GAG (FCG) scaffolds, two methods of fibrin incorporation into the CG matrix were investigated, drop-loading and injection loading, in order to assess the effect of each method on the resulting FCG microstructure (Fig. 1). Regardless of method, the crosslinked CG matrix was firstly soak-loaded with CaCl<sub>2</sub>, TBS and thrombin in the same concentrations and ratios as described above for the fibrin gels. Using the drop-loading technique, the fibrinogen was added on top of the CG matrix in a drop-wise fashion using a pipette. For the injection loading technique, the fibrinogen was injected into the matrix in a repeatable circular pattern using a 27 gauge needle. All scaffolds were left to polymerise at 37°C for 60 minutes. For the injection-loaded scaffolds, the optimum volume of fibrin to incorporate into the CG matrix was also assessed. Total volumes fibrin of 250µl, 300µl, 350µl and 400µl were injected into CG matrix (diameter 15.8mm, thickness 2mm). These volumes were based on preliminary experimental work that showed that volumes of greater than 400µl could not be maintained within each matrix.

#### 2.2 Characterisation of Fibrin-Collagen-Glycosaminoglycan Scaffold

#### 2.2.1 Histological staining of the FCG scaffold

Histological analysis was used to investigate fibrin infiltration into the CG matrix. Samples were formalin-fixed, embedded in Jung Tissue freezing medium (Laboratory Instruments and Supplies, Ashbourne, Ireland) and cryosectioned (Leica CM 1950, Wetzlar, Germany) to a thickness of 10µm, in both the transverse and longitudinal planes. Masson's Trichrome (Sigma-Aldrich) staining was performed using standard protocols by immersing in Bouin's solution, Biebrich scarlet-acid fuchsin, phosphotungstic/phosphomolybdic acid solution and aniline blue with the exclusion of haemotoxylin as no cells were present in these scaffolds. This process stained fibrin (red) and collagen (blue), allowing visualisation of both components in single sections. Samples were observed using a light microscope (ECLIPSE 90i; Nikon, Tokyo, Japan) and digital images were captured using the DS Ri camera and NIS elements software (Nikon).

#### 2.2.2 Microstructure of the FCG scaffold

The microstructure of the FCG scaffolds was examined using scanning electron microscopy (SEM) (Hitachi SU6600 VP-SEM; Hitachi High Technologies America Inc., Clarksburg, USA). To examine the infiltration of fibrin throughout the CG matrix, cross sections at different positions within the FCG scaffolds were formalin-fixed and critical point dried (Quorum E3000 CPD; Quorum Technologies, East Sussex, UK), fixed to an adhesive carbon stub, and then sputter coated with palladium/gold using a Polaron sputter coater (Quorum Technologies). CG matrix and fibrin gels prepared in an identical manner were used as references to compare structural features. SEM imaging was performed at an accelerating voltage of 15kV, utilising the secondary electron detector.

#### 2.2.3 Mechanical Characterisation of the FCG scaffold

Uniaxial compressive and tensile tests were performed to assess the effect of reinforcing fibrin with a crosslinked, acellular CG matrix. All samples were tested using a mechanical testing machine (Zwick/Roell, Ulm, Germany) fitted with a 5N load cell and were hydrated in PBS throughout the testing. Samples for compressive testing were as described in Section 2.1.2. For tensile testing, FCG scaffolds were produced in a dog-bone shape which were 63.5mm long with a narrowed centre of 3mm as per ASTM D638 (specimen type V)(ASTM D638 2014).. CG matrices and fibrin gels were also produced in this dog-bone shape as controls.

Compressive and tensile tests were conducted at a strain rate of 10% per minute as described by Haugh *et al* (Haugh et al. 2009). The modulus was defined as the slope of a linear fit to the stress–strain curve over 2–5% strain (Harley et al. 2007). Compressive and tensile testing was performed on three samples per scaffold type, each of which was tested in triplicate.

#### 2.3 Bioactivity Analysis

#### 2.3.1 Cell culture and seeding into the FCG scaffold

A human vSMC cell line was purchased from ATCC (CRL-1999; LGC Standards, Middlesex, UK). VSMCs are a contractile cell type, which allowed an appropriate assessment of the ability of the FCG scaffold to resist cellular-induced contraction. VSMCs were cultured using the recommended complete growth media consisting of Ham's F-12K (ATCC 30-2004, LGC Standards), supplemented with: 10% foetal bovine serum (Sigma-Aldrich), 2% penicillin/streptomycin (Sigma-Aldrich), 50µg/mL ascorbic acid (Sigma-Aldrich), 16µl/ml 1x ITS (Insulin, Transferrin, Selenium) (BD Biosciences, Oxford, UK), 10mM HEPES (4-(2hydroxyethyl)-1-piperazineethanesulfonic acid) (Sigma-Aldrich), 10mM TES (N-Tris(hydroxymethyl)methyl-2-aminoethanesulfonic acid) (Sigma-Aldrich) and 0.03mg/ml endothelial cell growth supplement (Sigma-Aldrich). Subculturing procedures were as per the manufacturer's instructions, using trypsin-EDTA solution (Sigma-Aldrich) for cell detachment. The culture medium was changed every 2 days and cell number was calculated using a haemocytometer.

For all tests, cells were seeded at a density of 1,000 cells/mm<sup>3</sup> per scaffold for both FCG and fibrin-only gels (Murphy & O'Brien 2010). Cells were initially suspended in the fibrinogen component of the fibrin and following injection, the FCG scaffolds were left to fully polymerise for 60 minutes at 37°C. To prevent adhesion to the cell culture plastic, scaffolds were then transferred to sterile 6 well plates and 4ml of media were added per well, together with  $20\mu g/ml$  of aprotinin (Sigma-Aldrich), a protease inhibitor (Ahmed et al. 2007). The culture medium was changed every 2-3 days and scaffolds were cultured for up to 7 days.

#### 2.3.2 Histological analysis of the cell distribution through the FCG scaffold over time

Histological analysis was performed to investigate cell dispersal in the FCG scaffold. Cellseeded scaffolds were analysed at 3, 5 and 7 days in both the transverse and longitudinal planes. Samples were sectioned as described earlier, and stained using Masson's Trichrome (Sigma-Aldrich) with the addition of Weigert's hematoxylin to counterstain the vSMCs (black). Digital images were captured using a light microscope (ECLIPSE 90i; Nikon) and the DS Ri camera and NIS elements software (Nikon).

#### 2.3.3 Cell Viability Analysis

To assess the cellular viability within scaffolds, staining was performed using a Live/Dead kit (Molecular Probes, Eugene, OR, USA) according to the manufacturer's protocol at 3, 5 and 7 days post-cell seeding. Calcein was used to stain viable cells green and Ethd-1 was used to stain dead cells red. Fluorescence microscopy was used to observe cell viability using the Leica DMIL Fluorescence microscope (Leica, Wetzlar, Germany).

#### 2.3.4 Cell Number Quantification

The Quant-iT PicoGreen dsDNA kit (BD BioSciences) was utilised according to the manufacturer's protocol to assess cell number within the FCG scaffold at days 0, 3, 5, and 7. Briefly, 100µl of the PicoGreen reagent solution was added to samples containing the 0.2M carbonate 1% TritonX (Fisher, Dublin, Ireland) cell lysate buffer and fluorescence was read at 538nm using a Varioskan Flash plate reader (ThermoScientific, Dreieich, Germany) and SkanIt RE for Varioskan software. Sample fluorescence was compared to a standard curve to determine cell number.

#### 2.4 Cell-mediated Contraction Analysis

The cell-mediated contraction of the fibrin and FCG scaffolds by the vSMCs was compared. The dimensions of both scaffold types were measured daily using Vernier callipers (Krunstoffwerke, Radionics, Dublin, Ireland) to assess scaffold contraction and graphed as percentage contraction versus day 0 (n=9). Thickness was also measured on day 0, 3, 5 and 7 using a micrometre. Scaffolds were unconstrained, and media was changed every 2-3 days.

#### 2.5 Statistical analysis

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Results are expressed as mean  $\pm$  standard deviation. Statistical significance was assessed using one-way analysis of variance (ANOVA), followed by Tukey post-hoc analysis. The sample size was n=3, except for the contraction assessment study, where n=9. P < 0.05 values were considered statistically significant throughout.

8

#### 3. Results

#### 3.1 Scaffold Fabrication and Characterisation

#### 3.1.1 Analysis of FCG scaffold microstructure

Histological analysis of the FCG scaffolds fabricated using drop-loading and injection loading techniques revealed that fibrin was not evenly dispersed throughout the CG scaffold in the drop-loaded FCG scaffolds. Instead, it was concentrated on the loaded side of the CG scaffold, as shown in Fig. 2A. This inconsistent incorporation in the drop-loaded group was also evident in the SEM images, as shown in Fig. 2B & C. In contrast, histological analysis of the scaffolds formed by injecting the fibrinogen component, showed excellent infiltration of fibrin throughout the CG matrix (Fig. 2D). Once more, the SEM images confirmed this, with a homogenous distribution of fibrin observed both in the transverse and longitudinal plane (Fig. 2E, F). The optimal FCG scaffold contained an infiltration ratio of 300µl of fibrin, loaded within a 15.8mm diameter, 2mm high CG scaffold (Fig. 2D). At this volume ratio, the fibrin component is seen throughout the thickness of the CG scaffold, unlike at the lower volumes, where insufficient incorporation was observed. Interestingly, with a larger volume above this level, the CG and fibrin were found to separate from each other (Fig. 2G where 350µl of fibrin was added).

#### 3.1.2 Mechanical Characterisation of the FCG scaffold

The compressive and tensile moduli of the FCG scaffold were found to be significantly higher than those of the fibrin-only gel (Fig. 3). Compressive testing of fibrin-only gels showed a compressive modulus of 0.49kPa  $\pm$  0.1, and when reinforced with the CG matrix this increased to 2.97kPa  $\pm$  0.5, demonstrating a six-fold increase in compressive modulus (Fig. 3A). The compressive modulus of the FCG scaffold was significantly lower (22% lower) than for the CG matrix (3.83kPa  $\pm$  0.4). The CG compressive modulus of elasticity presented here was stiffer than previously published CG data from this laboratory (Haugh et al. 2009), as a higher concentration of collagen was used. Tensile testing of fibrin-only gels showed a tensile modulus of 0.02MPa  $\pm$  0.01, and when reinforced with the CG matrix this increased to 0.6MPa  $\pm$  0.1, demonstrating a 30-fold increase in tensile modulus (Fig. 3B). The tensile modulus of the FCG scaffold was not significantly different to the CG-only matrix (0.59MPa  $\pm$  0.04).

#### 3.2 Bioactivity

#### 3.2.1 Histological analysis of cell distribution

VSMCs were successfully incorporated into the FCG scaffold, using the injection technique, with fibrin itself acting as the carrier material. To assess the distribution of cells throughout the FCG scaffold, Masson's Trichrome staining was completed on transverse and longitudinal sections at day 3 (Fig. 4A, B), day 5 and day 7 (Fig. 4C, D). This analysis showed cells to be distributed evenly both transversely and longitudinally throughout the FCG

scaffolds. No clumping of cells was observed, with cells contacting areas of both fibrin and collagen.

#### 3.2.2 Cell Viability Assessment

Having successfully infiltrated the FCG scaffold with cells, the viability of the cells was assessed using live/dead staining, which demonstrated that a high number of viable cells were observed at each time point (Fig. 5). No differences in viability were observed between cells in fibrin gels and in FCG scaffolds. Large numbers of live cells (green) were evident throughout all images and there were only a very small number of dead cells (red) (Fig. 5). Interestingly, there was also a noticeable increase in total live cell numbers over time, demonstrating the proliferation of the cells throughout the scaffold. At higher magnifications, a normal vSMC morphology was observed by day 7 and cells were seen to stretch around the CG pores (Fig. 5E). These results illustrate that the incorporation of fibrin into a CG matrix to form a FCG scaffold leads to an environment that supports cell viability and proliferation.

Cell proliferation over the 7 day culture period was quantified using a dsDNA PicoGreen assay (Fig. 6). A significant increase in dsDNA was observed at day 7 (compared to days 0 and 3 but not day 5) which confirmed the qualitative results seen in Fig. 5.

#### 3.3 Cell-mediated Contraction Study

Diameter and thickness was directly measured each day revealing unchanged dimensions in the FCG scaffold (Fig. 7) over the 7 day period. On the other hand, fibrin-only gels which were seeded using the same seeding density, and containing the same fibrinogen concentration, contracted to 30% of their original diameter after day 1 and to 10% by day 3 (Fig. 7A).

#### 4. Discussion

The overall aim of this study was to develop a fibrin-collagen-glycosaminoglycan (FCG) scaffold which would provide a dimensionally stable substrate for tissue engineering applications, by resisting cellular-induced contraction. Using an injection method to incorporate fibrin into a CG matrix ensured that a homogenous distribution of fibrin was achieved throughout the CG structure. Fibrin alone has a low tensile and compressive modulus; however, when reinforced with a CG matrix, the tensile and compressive modulus; however, when reinforced with a CG matrix, the tensile and proliferated within the FCG scaffold, which provided 100% resistance to cellular–induced contraction over 7 days. This resistance to offers convincing evidence that the FCG scaffold can provide a dimensionally stable system for tissue engineering applications. This fully natural scaffold containing fibrin, collagen and GAG offers many biological advantages for tissue engineering applications such as cartilage development and cardiovascular applications. The greatest potential for this material is perhaps in the cardiovascular field, with a particular focus on heart valves, where cell-mediated contraction of tissue-engineered devices has proven to be a problem in achieving functional repair tissue.

The use of an injection technique proved to be a suitable methodology for achieving a homogenous distribution of fibrin throughout the CG matrix, and this was demonstrated through histological analysis and SEM imaging. Uniformly infiltrating a porous natural polymer matrix with fibrin, before the fibrin polymerised posed a major challenge and drop-loading of fibrin onto the CG matrix proved unsuccessful, as a barrier of polymerised fibrin blocked the passage of the remaining fibrin through the CG structure. The volume ratio of fibrin to CG matrix was also assessed and a volume of 300µl of fibrin per 15.8mm diameter, 2mm high CG matrix, was found to be optimal. This represented approximately 0.8µl of fibrin per mm<sup>3</sup> of CG matrix. Interestingly, a higher volume ratio of fibrin to CG was more difficult to physically integrate into the CG matrix in the time prior to polymerisation.

Reinforcing fibrin with a CG matrix resulted in an increase in mechanical properties of the FCG scaffold. This increase in compressive and tensile moduli is explained by the framework of the CG struts within the FCG scaffold which reinforced the fibrin gel. The CG struts resist the applied load and result in a higher modulus for the FCG scaffold, than for fibrin gels. This was most obvious with regards to the tensile modulus, which increased 30-fold. There was no significant difference between the tensile modulus of the CG matrix and the FCG scaffolds. We propose that as fibrin gel has a low tensile modulus in comparison to the CG tensile modulus, it adds no resistance to alignment of the CG pores at under tension (Harley et al. 2007). However, the realignment of CG pores under compressive modulus for FCG than for CG matrix alone. The tensile strength of fibrin is difficult to measure due to its gel-like properties, thus it is not widely reported in the literature. In this study, all materials

were polymerised in the dogbone shape enabling the measurement of their tensile properties. Fibrin has previously been combined with polyglycolic acid, hyaluronic acid-tyramine and collagen (van Vlimmeren et al. 2013; Lee & Kurisawa 2013; Cummings et al. 2004; Rowe et al. 2007; Hokugo et al. 2006) and while these combinations increased the mechanical properties of fibrin, the resistance to cell-mediated contraction remained minimal. This was due perhaps to the bonding between the materials. However, it is not necessarily the case that the stiffer the scaffold the better, as matrix stiffness also has the ability to regulate cell motility, proliferation and differentiation in various cell types (Discher et al. 2005; Pho et al. 2008; Yip et al. 2009; Engler et al. 2006; Haugh et al. 2011). Additionally, many of these studies (van Vlimmeren et al. 2013; Lee & Kurisawa 2013; Cummings et al. 2004; Rowe et al. 2007; Hokugo et al. 2006) did not report strong cell viability.

Using collagen and GAG as reinforcement materials for a fibrin gel in this study has many advantages. Firstly, fibrin, collagen and GAGs are all natural materials which degrade enzymatically and the rate of degradation can be controlled through the use of protease inhibitors unlike synthetic materials which degrade in bulk (Shah et al. 2008). This correlation between degradation and tissue formation is important to ensure that the newly formed tissue does not contract. The Masson's Trichrome images here (Fig. 4) show that after 7 days in culture, both fibrin and CG are still present within the scaffold. As collagen is the building block for many biological tissues, it is advantageous to include it in a scaffold due to its ability to encourage normal biological functions in cells. Fibrin promotes ECM synthesis in the cells that it houses while also retaining deposited ECM due to the gel-like nature of the biomaterial. Fibrin, collagen and GAGs together have proven to be a suitable environment for both cardiovascular (Flanagan et al. 2006; van Vlimmeren et al. 2013; Alfonso et al. 2013; Neidert & Tranquillo 2006) and cartilage applications (Eyrich et al. 2007; Deponti et al. 2014). GAGs such as chondroitin sulphate and hyaluronic acid are found abundantly in the cardiac cushion, from which valvulogenesis occurs, and studies have shown that the signalling molecules contained within these GAGs, regulate further heart valve development (Armstrong & Bischoff 2004; Person et al. 2005; Eckert et al. 2012). When designing a scaffold for heart valve applications where the ability to repair and grow the tissue in vivo may be required, this can only be advantageous. GAGs are also present within the cartilage ECM, with chondroitin sulphate being the main component and hyaluronic acid, keratan sulphate and dermatan sulfate also present. Chondroitin sulphate has been shown to stimulate chondrogenesis in vitro and to promote cellular ingrowth and cartilaginous tissue formation in vivo (van Susante JLC et al. 2001; Buma et al. 2003).

When vSMCs were incorporated within the fibrin, an even distribution of cells throughout the FCG scaffold was achieved, as the fibrin polymerised quickly, securing the cells in position. This seeding process proved very efficient, demonstrated by the high cell retention levels seen. Notably, an initial homogenous cell distribution, as shown in this study, has

been linked with strong ECM production (Kim et al. 1998). In contrast, distribution and attachment of cells in many scaffolds require cells to migrate through from one side, which exposes the risk of low cell attachment, making the scaffolds unsuitable for tissue development (Zund et al. 1999; Ye, Zünd, Jockenhoevel, et al. 2000). As demonstrated through live/dead staining and DNA quantification, the increase in cell numbers over the 7 days showed proliferation within the scaffold, leading to a high cell density. The reduction in cell number between days 5 and 7 in this scaffold is not statistically significant, and should be viewed as normal experimental variation. While there are some dead cells observed by day 7 in Fig. 5(D and E), overall cell viability is high (Fig. 5) with a statistical increase in cell number overall (Fig. 6). Together, these findings have positive clinical implications, as the efficient seeding process, and the ability of the FCG scaffold to support proliferation, would ensure fewer donor cells were required prior to the construction of regenerated tissue.

Fibrin has many uses as a tissue engineering scaffold, however many in vivo studies have shown a reduction in function resulting from the susceptibility of fibrin to cell-mediated contraction. Over seven days in culture, the FCG scaffold developed in this study resisted contraction by vSMCs, thereby achieving the primary aim of the study. Three separate factors may have influenced the ability of the FCG scaffold to resist contraction. Firstly, increased stiffness of the substrate, due to crosslinking, has been reported to reduce cellmediated contraction (Syedain et al. 2009; Sheu et al. 2001). FCG is a stiffer material than fibrin alone and so its resistance to deformation and bending is higher. The crosslinking of the CG matrix ensures that this stiffness is representative of the microstructure of the matrix and not a bulk material property. This ensures that at a size scale relevant to the cells encapsulated, the microstructure of CG struts resists the cell-mediated contraction. Valvular interstitial cells (VICs) have demonstrated the ability to sense their biomechanical environment resulting in pathological differentiation in vitro to osteoblast lineages. This has been attributed to a combination of both culturing on matrices with a compressive stiffness of higher than ~110kPa and biochemical changes resulting in an increase in cell stiffness (Wyss et al. 2012; Yip et al. 2009; Merryman et al. 2007). This suggests that while the FCG scaffold is stiff enough to resist cellular-induced contraction, this stiffness will not cause calcification of cells. In fact, native porcine aortic heart valves have a reported stiffness of 703kPa and have an ultimate tensile strength of 1450kPa (Merryman, Huang, et al. 2006; Merryman, Engelmayr, et al. 2006). Secondly, an increase in collagen density has been reported to reduce cell-mediated contraction (Legant et al. 2009). The concentrations of both collagen and GAG utilised in this study were optimised to provide the most stable combination of each macromolecule. 0.75% collagen w/v is a higher concentration than used for orthopaedic applications in our lab (Tierney et al. 2009; Keogh et al. 2010), however, once infiltrated with fibrin, other concentrations did not have the inherent structural stability, even when crosslinked, to endure handling and manipulation when infiltrated with fibrin during experiments. Thirdly, GAGs have also been shown to reduce cell-mediated contraction (van Vlimmeren et al. 2013). In studies where minimising

contraction of fibrin was examined, the amount of GAGs available to the tissue had a direct impact on the level of cell contraction that occurred, with greater amounts of GAG leading to lower cell contraction. This was hypothesised to be due to the damping effect of the GAGs, however the types of GAG in the study were not determined (van Vlimmeren et al. 2013). From a heart valve perspective, the spongiosa layer of the heart valve, which is comprised primarily of GAGs, works as a cushion allowing the spongiosa and ventricularis to shear relative to each other and the water retained by the GAGs also provides a damping effect to the tissue (Misfeld & Sievers 2007). Findings from the static analysis suggest that when exposed to a dynamic environment, these materials will continue to demonstrate appropriate dimensional stability and resistance to proteolytic activity and haemodynamic forces. This clearly warrants further investigation.

Taken together, the results herein provide compelling evidence that the FCG scaffold developed here provides a natural scaffold which can resist cell-mediated contraction normally seen in fibrin scaffolds. The structure of the FCG scaffold is homogenous and provides higher compressive and tensile moduli than fibrin alone. Cells can be incorporated efficiently providing an excellent cell distribution of viable cells which proliferate over time. Over 7 days in unconstrained culture, the dimensions of the FCG scaffold did not change, showing an ability to resist cellular-induced contraction, while still maintaining cell viability. As collagen also forms the main component of a range of other tissue types within the body, the FCG scaffold presented has potential for applications in areas such as cartilage repair, wound healing and in applications where angiogenesis is important (Weinandy et al. 2014) due to its mechanical properties and unique biological framework. We are currently focussing on cardiovascular applications for this material, specifically as a heart valve scaffold.

#### 5. Conclusion

This study presents the development of a natural fibrin-collagen-glycosaminoglycan scaffold with unique mechanical and biological properties. Using a technique where fibrin is injected into a collagen-glycosaminoglycan porous matrix, an effective manufacturing process that ensured excellent integration of all scaffold constituents was established. The compressive and tensile moduli of this FCG scaffold were significantly increased, in comparison to fibrin alone. Consistent cell viability, with an increase in VSMC proliferation over 7 days, was also demonstrated. Crucially, the ability of this FCG scaffold to resist cell-mediated contraction was demonstrated, with no change in scaffold dimensions over 7 days. These results show the potential of this material for use in applications where dimensional stability is crucial to the functionality of the tissue and also where a solution comprised of all natural materials makes a more attractive option than currently available materials.

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Acceleration

### **Figure Captions**

**Figure 1: Methods of producing fibrin-collagen-glycosaminoglycan (FCG) scaffolds.** The CG structure is a freeze-dried crosslinked matrix to which fibrin is added. For both the drop-loading or injection method the first three fibrin ingredients (CaCl<sub>2</sub>, thrombin and TBS) are first added to the CG matrix. For the drop-loading method, fibrinogen is added in a second step by being dropped on top of the matrix. For the injection method, fibrinogen (& cells) is injected using a 27 gauge syringe into the CG matrix, in a circular pattern. The injection method produced more consistent FCG scaffolds.

**Figure 2:** The injection method of fibrin incorporation showed a more homogenous fibrin distribution through the Collagen-Glycosaminoglycan (CG) matrix than the drop loading method. (A) Masson's Trichrome (MT) staining of drop-loaded scaffolds showed uneven infiltration of fibrin (red) through the collagen glycosaminoglycan (CG) matrix (blue). (B) Scanning Electron Microscopy (SEM) confirmed this, revealing little fibrin evident within the CG matrix. (C) MT staining of injection loaded FCG scaffolds show fibrin (300µl) throughout the CG matrix. (D) SEM images of injection loaded scaffolds show fibrin and CG intermingled. (E) MT image where 350µl of fibrin was injection loaded through the CG structure resulting in overloading of the CG structure with fibrin as seen by the large amount of fibrin which remained on top of the CG structure. (N=3 in all cases).

**Figure 3: The compressive and tensile moduli of fibrin were significantly increased when fibrin was reinforced with a crosslinked collagen glycosaminoglycan (CG) matrix.** (A) Compressive modulus of fibrin is significantly increased when incorporated into a CG structure. The compressive modulus of CG is significantly decreased when fibrin is incorporated. (B) The tensile modulus of CG and FCG were not significantly different from each other, but both were independently significant to fibrin alone. N=3, for significant difference, p<0.05 (\*).

Figure 4: Masson's Trichrome staining of fibrin-collagen-glycosaminoglycan (FCG) scaffolds, taken at day 3 (A & B) and day 7 (C & D) that show fibrin (red), collagen (blue) and cells (black) throughout the entire scaffold in both the longitudinal (A & C) and transverse (B & D) planes. Images taken at day 5 are not shown here, but show the same detail. Scale bar represents 100µm. N=3.

Figure 5: Live/dead staining of vascular smooth muscle cells (vSMC) in a fibrin-collagenglycosaminoglycan (FCG) scaffold over 7 days, show excellent viability with very few dead cells (red) present. Live cells fluoresce green and are seen to increase in number from day 3 (A & B) to days 5 (C & D) and 7 (E & F). All samples were seeded at the same seeding density of 1,000 cells per

mm<sup>3</sup> of scaffold. On day 3, excellent viability is seen with space remaining within the scaffold structure. By Day 5 the cell number has increased, with very few dead cells evident. Day 7 showed scaffolds with confluent cell populations. As these samples were 2mm deep, the images are picking up cells, both on the surface of the scaffold and cells within the scaffold. The deeper cells are out of focus in the images. N=3 at each time point.

Figure 6: An increase in dsDNA is seen over the timepoints using PicoGreen assay, indicating that the vascular smooth muscle cells are proliferating in the fibrin-collagen-glycosaminoglycan (FCG) scaffold. Based on p<0.05 cell number was shown to be significantly higher at Day 5 than Day 0 and Day 3, and higher at Day 7 than Day 0 (\*\*\*). These results indicate that proliferation of the cells was occurring in the FCG scaffold. Results represent three independent samples tested in triplicate.

**Figure 7: Fibrin-collagen-glycosaminoglycan (FCG) scaffolds resisted cell-mediated contraction in vitro, by vascular smooth muscle cells up to 7 days.** As shown in (A), fibrin gels with the same seeding density had contracted to approximately 10% of their original diameter after 3 days. Graph shows mean ± standard deviation. N=9 for these tests. (B) Macro image of cell seeded fibrin gels at day 0 and subsequently following contraction, at day 7. (C) Macro image of FCG scaffold at day 0, and (D) at day 7 where no contraction of the FCG scaffold had occurred.

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# **Cell number based on DNA flourescence** \*\*\* % of original cell number 200 150<sup>-</sup> 100 **50** 0 210

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